

# A novel ultra-low temperature cofired Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> ceramic and its chemical compatibility with metal electrodes

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Abstract Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> ceramic was investigated as a promising microwave dielectric material for the ultra-lowtemperature co-fired ceramic (ULTCC) technology. Dense Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> ceramic was prepared using the conventional solid-state method from 560 to 640 °C. X-ray diffraction data show that Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> ceramic crystallized into a cubic garnet structure with a space group Ia-3d. The sample sintered at 600 °C for 4 h has the highest relative density of 96.3 % and exhibits the optimum microwave properties with a relative permittivity of 22.3, a quality factor of 19,960 GHz (at 8.7 GHz), and a temperature coefficient of resonance frequency of +15.5 ppm/°C. The Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> ceramic was found to be chemically compatible with highly conductive aluminum and sliver electrode. These results confirm that Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> ceramic can be a promising candidate for the ULTCC technology.

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### 1 Introduction

To achieve the miniaturization and integration of the microwave components for wireless communication, low-temperature co-fired ceramic (LTCC) technology has become an important method that enables the fabrication of three-dimensional ceramic modules with a low dielectric loss and co-fired metal electrodes [1, 2]. For LTCC technology, a low sintering temperature lower than the melting point of metal electrodes (961 °C for Ag) is critical in addition to the appropriate relative permittivity ( $\varepsilon_r$ ), a high quality factor ( $Q \times f$ ), and a near-zero temperature coefficient of resonant frequency ( $\tau_f$ ) [3–5].

Recently, searching for novel microwave dielectric ceramics with intrinsic low firing temperatures, such as TeO<sub>2</sub>-based [6, 7], Bi<sub>2</sub>O<sub>3</sub>-based [8, 9], and MoO<sub>3</sub>-based [10] systems, has attracted much attention. Some of them could co-fire with aluminum electrodes due to their ultra-low sintering temperatures <660 °C. The application of Al as the inner electrodes has accelerated the ultra-low-temperature co-fired ceramic (ULTCC) technology [10, 11]. More recently, several ULTCCs have been reported by the researchers, for example, BaTe<sub>4</sub>O<sub>9</sub> ( $\varepsilon_r = 17.5$ ,  $Q \times f = 54,700$  GHz,  $\tau_f = -90$  ppm/°C and S. T. = 550 °C) [6], Bi<sub>2</sub>Mo<sub>2</sub>O<sub>9</sub> ( $\varepsilon_r = 38$ ,  $Q \times f = 12,500$  GHz,  $\tau_f = +31$  ppm/°C and S. T. = 620 °C) [10], and NaAgMoO<sub>4</sub> ( $\varepsilon_r = 7.9$ ,  $Q \times f = 33,000$  GHz,  $\tau_f = -120$  ppm/°C and S. T. = 400 °C) [12].

In our previous work [13, 14], some garnet vanadates were reported to have good microwave dielectric properties, such as, LiCa<sub>3</sub>ZnV<sub>3</sub>O<sub>12</sub> ( $\varepsilon_r \sim 11.5$ ,  $Q \times f \sim 81,100$  GHz,  $\tau_f \sim -72$  ppm/°C, S. T.  $\sim 900$  °C) and Na<sub>2</sub>YMg<sub>2</sub>V<sub>3</sub>O<sub>12</sub> ( $\varepsilon_r \sim 12.3$ ,  $Q \times f \sim 23,180$  GHz and  $\tau_f \sim -4.1$  ppm/°C, S. T.  $\sim 850$  °C). Most of them have chemical compatibility with silver electrode when sintered at their densification temperatures, making them possible candidates for



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LTCC applications. More recently, Zhou et al. [15] reported that Na<sub>2</sub>BiMg<sub>2</sub>V<sub>3</sub>O<sub>12</sub> has an ultra-low sintering temperature  $\sim\!660$  °C with a  $\varepsilon_r\sim23.2$ , a  $Q\times f\sim3700$  GHz and a near-zero  $\tau_f\sim+8.2$  ppm/°C. Therefore, it is worthwhile to investigate the Na<sub>2</sub>BiMV<sub>3</sub>O<sub>12</sub> (M = Zn²+, Co²+, Ni²+) systems for an attempt to search for novel ultra-low-temperature co-fired ceramics.

In the present paper, an ultra-low-temperature co-fired ceramic Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> with garnet structure was reported. The sintering behavior, microstructure, microwave dielectric properties, and its chemical compatibility with both aluminum and silver were investigated in detail.

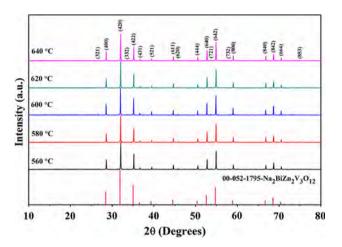


Fig. 1 X-ray diffraction patterns of Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> ceramics sintered at different temperatures from 560 to 640  $^{\circ}C$ 

# 2 Experimental procedure

Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> ceramic was prepared by the conventional solid-state reaction with high-purity oxides or carbonate powders, Na<sub>2</sub>CO<sub>3</sub> (99 %, Guo-Yao Co. Ltd., Shanghai, China), Bi<sub>2</sub>O<sub>3</sub> (99 %, West Long Chemical Co., Ltd., Guangdong, China), ZnO (99 %, Guo-Yao Co. Ltd., Shanghai, China), and NH<sub>4</sub>VO<sub>3</sub> (>99 %, West Long Chemical Co., Ltd., Guangdong, China). Raw materials were weighed stoichiometrically and mixed, ball-milled in alcohol media for 6 h, followed by the calcination at 520 °C for 4 h. The calcined powders were ball-milled for 6 h, dried, and pressed into cylinders with 12 mm in diameter and 7 mm in height under a pressure of 200 MPa. Polyvinyl alcohol (PVA) was added to the powders as binder. The samples were fired at 500 °C for 2 h to burnout the organic binder, and then sintered at 560-640 °C for 4 h with a heating rate of 5 °C/min. To investigate the chemical compatibility, Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> powders were mixed with 20 wt% aluminum and silver powders and cofired at 640 °C for 4 h.

X-ray diffraction (XRD) was employed to analyze the phase composition (1.54059 Å, Model X'Pert PRO, PANalytical, Almelo, Holland) in the  $2\theta$  range of 10– $80^{\circ}$ . Bulk densities of the sintered samples were measured using Archimede's method. The microstructures were examined by scanning electron microscopy (SEM; FE-SEM, Model S4800, Hitachi, Japan). The microwave dielectric properties were analyzed using a network analyzer (Model

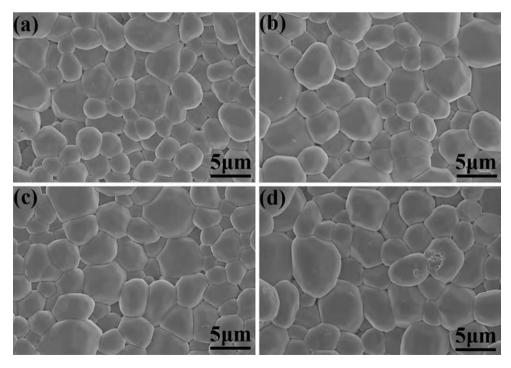


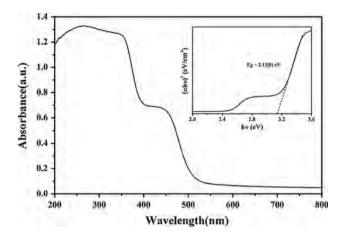
Fig. 2 SEM images of Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> ceramics sintered at a 560 °C, b 580 °C, c 600 °C and d 620 °C for 4 h in air



N5230A, Agilent Co., Palo Alto, California) and a temperature chamber (Delta 9039, Delta Design, San Diego, CA). The  $\tau_f$  value was calculated using the following relationship:

$$\tau_f = \frac{f_{85} - f_{25}}{(85 - 25) \times f_{25}} \tag{1}$$

where,  $f_{85}$  and  $f_{25}$  are the resonant frequencies of the dielectric resonator at temperature 85 and 25 °C, respectively.



**Fig. 3** UV–Vis light absorption spectrum of the  $Na_2BiZn_2V_3O_{12}$ . *Inset* plots of  $(\alpha h \nu)^2$  versus energy  $h \nu$  of  $Na_2BiZn_2V_3O_{12}$  ceramic

# **Fig. 4** The relative densities (a) and microwave dielectric properties $\varepsilon_r$ (b), $Q \times f$ (c), and $\tau_f$ (d) of Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> ceramics at different sintering temperatures

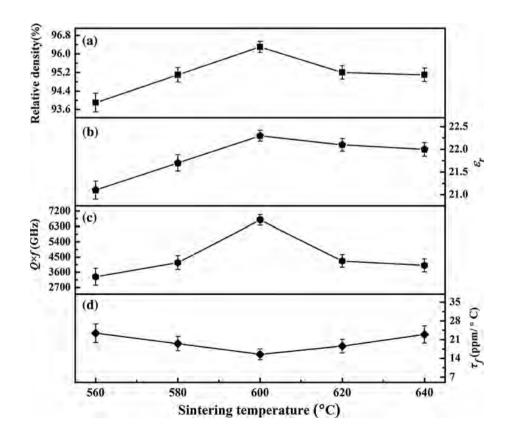
#### 3 Results and discussions

Figure 1 shows the XRD patterns of the  $Na_2BiZn_2V_3O_{12}$  ceramics sintered from 560 to 640 °C for 4 h. The observed peaks matched well with JCPDS card No. 52-1795 for  $Na_2BiZn_2V_3O_{12}$  with no secondary phases detected, indicating the formation of pure-phase  $Na_2-BiZn_2V_3O_{12}$  with a cubic garnet structure.

SEM micrographs of  $Na_2BiZn_2V_3O_{12}$  ceramics sintered at different temperatures are shown in Fig. 2. It shows that  $Na_2BiZn_2V_3O_{12}$  ceramics could be well densified within the certain temperature range of 560–620 °C. The ceramics sintered at 560 °C showed a relatively porous microstructure (Fig. 2a) with small grains about 2–3  $\mu$ m. With the increasing sintering temperature, the grain size increased along with a significant decrease in the porosity. A uniform and dense microstructure with closely packed grain morphology ( $\sim$ 5  $\mu$ m in average grain size) was obtained in the sample sintered at 600 °C. However, as the sintering temperature increased to 620 °C, abnormal grain growth and grain melting began to appear.

The optical absorption properties  $Na_2BiZn_2V_3O_{12}$  ceramic were investigated by UV–Vis techniques. The values of the band gap energy ( $E_g$ ) were calculated using following equation [16–18]:

$$(\alpha h v) = A \left( h v - E_g \right)^n \tag{2}$$





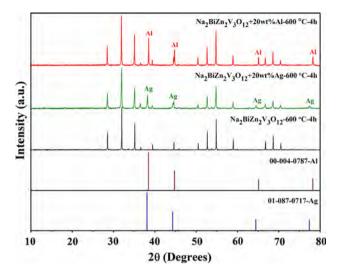


Fig. 5 X-ray diffraction patterns of  $Na_2BiZn_2V_3O_{12}$  cofired ceramics with 20 wt% Al and 20 wt% Ag at 600 °C for 4 h

where A is a proportional constant, h is Planck's constant, v is the frequency of vibration,  $E_g$  is the band gap energy,  $\alpha$  is the absorption coefficient per unit length, and n is 0.5 and 2.0 for a direct transition semiconductor and indirect transition semiconductor, respectively [19–21]. The Ultraviolet–visible diffuse reflection spectra and plots of  $(\alpha hv)^2$  versus energy hv of Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> ceramic is displayed in Fig. 3. In the inset, the Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> sample shows a band gap energy of 3.13(8) eV.

Figure 4 shows the variations in relative densities and microwave dielectric properties of Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> ceramics as a function of the sintering temperature. The relative density showed a obvious dependence on the sintering temperature and a maximum value of 96.3 % (4.73 g/cm<sup>3</sup>) of the theoretical density  $\sim$ 4.91 g/cm<sup>3</sup>) at 600 °C. As shown in Fig. 4b,  $\varepsilon_r$  increased from 21.1 to 22.3 as the sintering temperature increased from 560 to 600 °C,

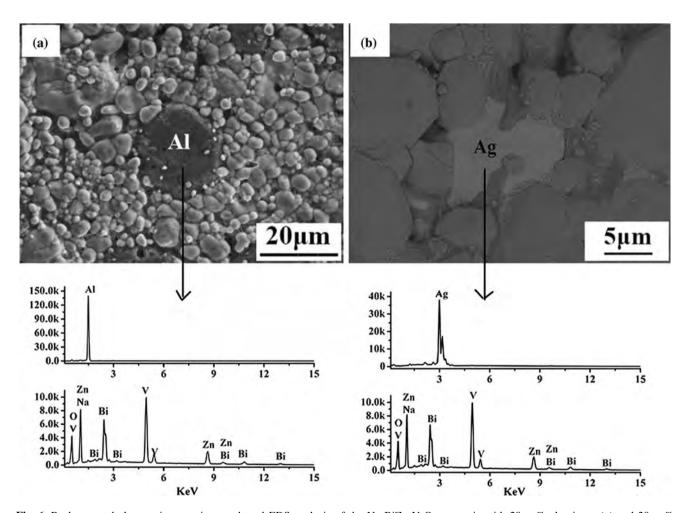


Fig. 6 Backscattered electron image micrograph and EDS analysis of the  $Na_2BiZn_2V_3O_{12}$  ceramic with 20 wt% aluminum (a) and 20 wt% silver (b) sintered at 600 °C for 4 h



and then slightly decreased with further increasing temperature. The variation in  $\varepsilon_r$  with the increasing sintering temperature is consistent with that of the relative density. The lower permittivity at lower sintering temperature could be partly attributed to the existence of pores. The influence of the porosity on  $\varepsilon_r$  could be eliminated by applying Bosman and Having's correction [22, 23]:

$$\varepsilon_{\text{corrected}} = \varepsilon_m (1 + 1.5p)$$
 (3)

where, p is the fractional porosity;  $\varepsilon_{\rm corrected}$  and  $\varepsilon_m$  are the corrected and measured values of permittivity, respectively. The  $\varepsilon_{\rm corrected}$  is about 23.5 for Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> sintered at 600 °C.

It is well known that there are many factors contributing to the dielectric loss at microwave region: the intrinsic factors and the extrinsic ones such as impurities, substitution, grain boundaries, grain morphology and shape, secondary phase, pores, dominate the  $Q \times f$  value [24, 25]. As shown in Fig. 4c, an increase in  $Q \times f$  value with sintering temperature was observed and a maximum value of 19,960 GHz was reached when sintered at 600 °C for 4 h. Thereafter, the  $Q \times f$  value decreased, which might be due to extrinsic factors, such as the increase of pores and the abnormal grain growth. The  $\tau_f$  values of Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> ceramics slightly fluctuated around +18 ppm/°C over the sintering range from 560 to 600 °C.

The XRD patterns of the cofired samples with 20 wt% aluminum and silver sintered at 600 °C are shown in Fig. 5 and XRD pattern of the pure Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> ceramic is also presented for comparison. For the cofired ceramic samples, only the peaks of Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> and the metals could be observed with no additional peaks detected. The backscattered electron image micrograph and EDS analysis of the cofired ceramics with 20 wt% aluminum (a) and silver (b) are shown in Fig. 6. The analysis revealed that the cofired ceramics were composed of both Na<sub>2</sub>BiZn<sub>2</sub>-V<sub>3</sub>O<sub>12</sub> grains and metal grains. These results confirm no chemical reaction between Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> and aluminum or silver when sintered at 600 °C for 4 h.

# 4 Conclusions

Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> ceramic can be prepared by conventional solid state reaction method and densified after sintering above 560 °C for 4 h in air. Optical absorption properties were investigated by UV–Vis techniques. The best microwave dielectric properties can be obtained in Na<sub>2</sub>BiZn<sub>2</sub>-V<sub>3</sub>O<sub>12</sub> ceramic sintered at 600 °C for 4 h, with a permittivity of 22.3,  $Q \times f$  value of 19,960 GHz (at 8.7 GHz), and a positive  $\tau_f$  value of +15.5 ppm/°C. From the XRD and EDS analysis, the Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> ceramic was found to be chemically compatible with aluminum or

silver powders at its sintering temperatures. Based on the experimental results of this research, Na<sub>2</sub>BiZn<sub>2</sub>V<sub>3</sub>O<sub>12</sub> seems to be an attractive candidate for the ultra-low temperature co-fired ceramic technology.

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